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RESEARCH ARTICLE

STUDY OF OXIDATION BEHAVIOR OF LEAD THIN FILMS BY THERMAL EVAPORATION METHOD

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ABSTRACT

The lead thin films were deposited onto amorphous glass substrate using a vacuum thermal deposition technique by varying annealing time to study its oxidation behaviour. The XRD pattern clearly exhibited the phase changes in accordance with the annealing time and reaches the menopause of PbO at a maximum annealing time. The surface morphology, particle size and uniformity of the films were ascribed from SEM micrograms. The change in surface morphology or topography with the function of oxidation time was observed from amplitude and functional parameters. The optical UV-Vis spectra showed that the directly allowed transitions band gap values were decreased due to the rise of annealing time. The functional groups were analysed by using FT-IR techniques.

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INTRODUCTION

Over the last few decades, research interest in the study of nanomaterials and their applications has been increasing, since these materials often demonstrate very different properties at the nanoscale level as compared to those at the macro level, such as new optical, magnetic, and electronic characteristics (Rosenqvist *et al.*, 2003; Cha *et al.*, 2008; Chaudhry *et al.*, 2009; Baker *et al.*, 2002). Furthermore, nanomaterials with high aspect-ratio structures and large surface areas offer exciting research possibilities because of such novel physical or chemical properties. As a result, the synthesis and characterization of one and two-dimensional metal oxide nanostructures have attracted considerable attention among the researchers. The different phases (α , β) and mixture of lead oxide films such as PbO, Pb₂O, PbO₂, Pb₂O₃ and Pb₃O₄ are (IV-VI group) semiconductor materials narrow band gap (1.28eV–2.80eV) (Droessler *et al.*, 2012) which are found useful in photovoltaic materials. A number of theoretical and experimental studies have shown an interesting optical and electronic property of lead oxide and suggesting it to be a very useful photovoltaic material with strong absorption cross sections, high carrier mobility and conductivity (Schottmiller, 1966). The nature of the band gap is suitable for photovoltaic materials design, however information on lead oxides is somewhat limited and at times contradictory requiring further study (Sharon *et al.*, 1998; Thangaraju, 2000; Veluchamy and Minoura, 1994). However, few studies have been conducted on the fabrication of lead oxide by vacuum thermal evaporation

because of the lack of practical preparation methods for such materials (Droessler *et al.*, 2012). Thermal evaporation is one of the most widely used techniques of physical vapor deposition (PVD). The deposited coatings or films by this technique are usually have a thickness in the range of angstroms to microns and are composed of a single material or layers of multiple materials. The coating materials can be either molecules, including nitrides and oxides, or pure atomic elements such as both metals and non-metals. The evaporation system design allows to adjust a number of parameters to obtain desired results for variables such as grain structure, uniformity, thickness, stress, adhesion strength, optical and electrical properties.

In the present study, a systematic investigation of lead thin films, oxidized at different annealing time, shows a remarkable difference in structural, surface morphological and optical properties.

MATERIALS AND METHODS

Thin films of Lead (Pb) were deposited onto a glass substrate by vacuum thermal evaporation method. The lead thin films were obtained by evaporating commercial high purity lead (Pb) powder (99.9% Aldrich) from an electrically heated molybdenum boat at about 4×10^{-5} Torr. Initially the glass substrate was heated using a substrate heater at 373K. The deposited Pb films were oxidized at 543 K by varying annealing time. The important step before the deposition of films is the careful cleaning of the substrates. As the cleanliness of the substrate has a direct bearing on the

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adherence of the film, the following procedure was adopted for cleaning the substrates. The slides were washed with liquid detergent and then boiled in concentrated chromic acid (0.5 M) for 2 hours. After which, the slides were kept in the same solution for further 48 hours. The substrates were then washed with double distilled water, and then cleaned by ultrasonication and de-ionized water. Finally, the substrates were cleaned using acetone (AR grade).

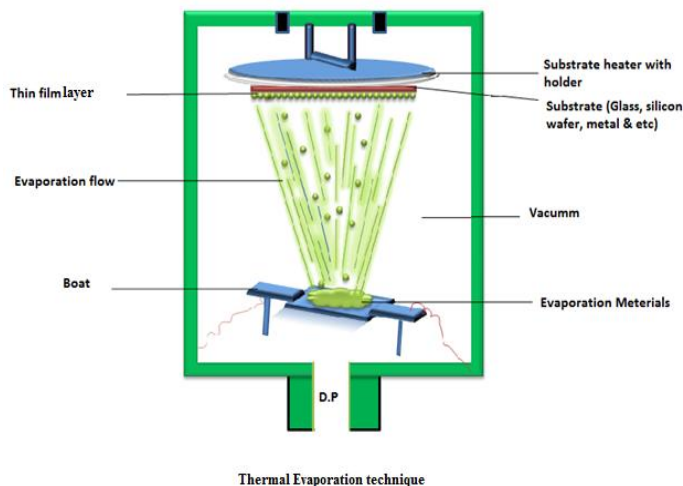


Fig. 1. Schematic diagram of thermal evaporation

Fig 1. shows the schematic diagram of thermal evaporation technique. The deposited Pb films were annealed at 543K 30 min, 60 min, 120 min, and 180 min and then characterized.

Characterizations

The X-Ray Diffraction (XRD) patterns were obtained for the prepared lead oxide thin films with X’Pert PRO Diffractometer using CuK α 1 radiation ($\lambda=1.54060 \text{ \AA}$) with operating voltage 15 KV and a current of 30 mA. Scanning electron micrographs were obtained with HITACHI S-3000 N. Topography or surface morphology of lead oxide thin films was characterized using a Park Systems XE70 instrument, and it was conducted in the ambient atmosphere in non-contact mode. The scan resolution is 64 pixels x 64 pixels, set point 0.7 μ m and scan rate 0.5 Hz. Optical transmission spectra were taken in the wavelength range 300 – 800 nm using Shimadzu UV-Visible 1800 double beam spectrophotometer. The infrared transmittance spectra of the prepared samples were measured at room temperature in the range 1200–400 cm^{-1} using a recording PerkinElmer FTIR spectrophotometer. Thickness of the films was obtained by Stylus Profilometer Mitutoyo Surf test SJ – 301.

RESULTS AND DISCUSSION

X-Ray diffraction

The X-Ray Diffraction (XRD) pattern of lead oxide thin films, compounds with the function of annealing time was shown in Fig. 2.

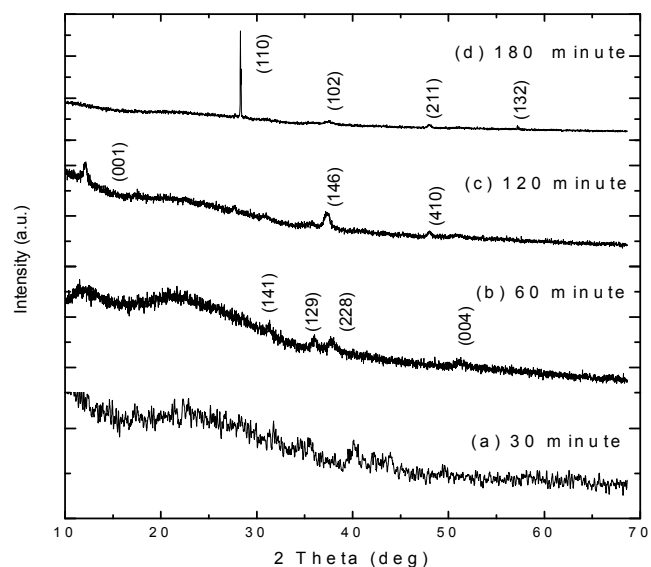


Figure 2. X-ray diffraction pattern of lead films oxidized at 543K for different annealing time

The lower annealing time respective thin film has no diffraction peaks which mean that the films is in amorphous nature. The diffraction patterns corresponding for 60 min and 120 min reveal that mixed compounds of lead oxides (Pb_3O_4 and Pb_2O_3). The monophasic lead oxide thin films was only achieved in 180 min. The mixed compounds of lead and oxide in the thin films at lower annealing times react with each other and form the mono phase of lead oxide at a higher annealing time of 180 min. Fig 2. (d) shows four diffraction peaks along the miller planes (110), (102), (211) and (132) at 2θ values of 31.8° , 42.9° , 55.23° and 66.48° . These diffraction peaks were matched with standard JCPDS data of lead oxide.

Table 1. Comparisons of the results from XRD analysis of the Lead oxide thin films

Annealing time (min)	Position (2θ) Standard	(deg) Observed	JCPDS Card No	Stoichiometry	System
30	-	-	-	-	-
60	35.46	35.44	761831	$\text{Pb}_2\text{O}_{3.33}$	Monoclinic
	41.06	41.09	761831	$\text{Pb}_2\text{O}_{3.33}$	Monoclinic
	43.04	43.10	761831	$\text{Pb}_2\text{O}_{3.33}$	Monoclinic
	56.03	56.01	761831	$\text{Pb}_2\text{O}_{3.33}$	Monoclinic
120	12.55	12.68	230331	Pb_2O_3	Monoclinic
	42.44	42.42	730532	Pb_3O_4	Tetragonal
	55.36	55.38	710562	Pb_3O_4	Orthorhombic
180	31.80	31.83	851739	PbO	Tetragonal
	42.85	42.96	850711	PbO	Tetragonal
	55.15	55.23	850711	PbO	Tetragonal
	66.30	66.48	851289	PbO	Tetragonal

The crystallite sizes were estimated by applying Debye Scherrer’s formula (Cullity, 1956).

$$D = 0.94\lambda/\beta\cos\theta \dots\dots\dots (1)$$

Where D is the grain size, λ is the X-ray wavelength corresponding to CuK α radiation ($\lambda= 1.54060\text{\AA}$), β is the full-width at half-maximum (FWHM) of x-ray diffraction pattern peaks and θ is the Bragg angle.

The dislocation density δ , which represents the amount of defects in the film, was determined from the formula (Beltran *et al.*, 2003),

$$\delta = 1/D^2 \quad \dots\dots\dots(2)$$

The micro strain (ϵ) was calculated by the formula (Bedir *et al.*, 2005),

$$\epsilon = \beta \cos\theta/4 \quad \dots\dots\dots(3)$$

The thicknesses (t) of the as-prepared thin films are increased with increasing annealing time, which means that the rise in particle size is due to the merging of smaller crystals (Peter 1979).

The number of crystallites per unit area (N) of the films is estimated using the relation (Ning *et al.*, 2006).

$$N = \frac{t}{D^3} / \text{unit area} \quad \dots\dots\dots(4)$$

The calculated values of crystallite size, micro strain, dislocation density and unit area crystallite are given in Table 1.2.

Table 1.2. Grain size, Micro strain and Dislocation density of lead oxide thin films

Annealing time (min)	grain size D (nm)	Micro Strain ($\epsilon \times 10^{-3}$) (lines ² m ⁻⁴)	Dislocation Density $\delta \times 10^{14}$	Unit area $N = t/D^3 \times 10^{14}$	Thickness of the film (μm)
30	-	-	-	-	0.87
60	88.06	13.85	1.02	11.46	0.91
120	98.63	24.79	1.49	13.32	1.10
180	162.58	17.99	1.83	3.44	1.48

3. Scanning Electron Microscopy (SEM)

The Scanning electron microscopy (SEM) image of lead oxide thin films as a function of annealing time was shown in Fig. 3.1 – 3.4. The SEM image clearly shows that the particle size and crystallinity increase with increase in annealing time. The thin film annealed for 180 min shows the higher crystalline nature.

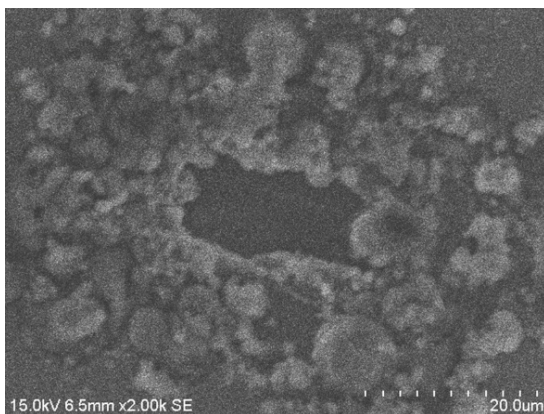


Figure 3.1. SEM images of lead films annealed at 543K, for 30 min

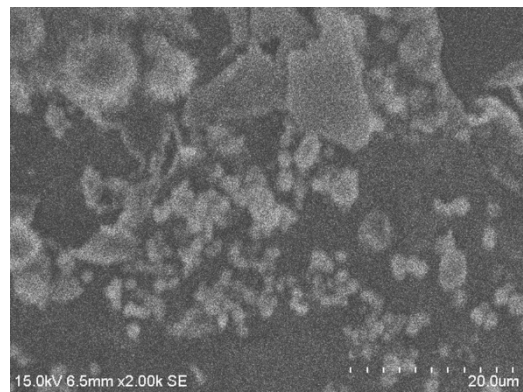


Figure 3.2. SEM images of lead films annealed at 543K, for 60 min

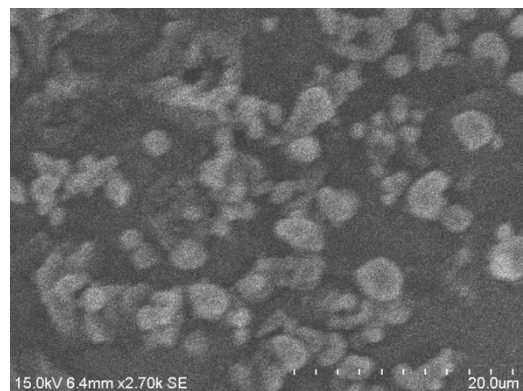


Figure 3.3. SEM images of lead films annealed at 543K, for 120 min

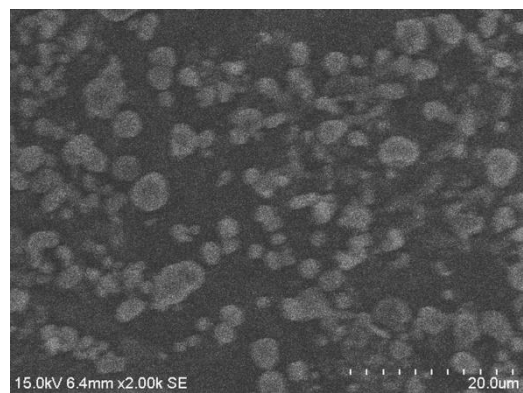


Figure 3.4. SEM images of lead films annealed at 543K, for 180 min

The Fig 3.1 – 3.4 revealed that the films are coarse with an irregular shape. It is indicated that the morphology of the lead oxide thin films was affected by thermal oxidation treatment. The grain boundaries of lead oxide thin films become nanoscale range with particle size seem to be varying from other samples with gaps in between and there is a change in morphology or clearly seen grain boundaries were observed. The creation of larger grains was due to the coalescence of small grains into bigger one, which can be seen from 180 min sample (Family and Vicsek, 1991; Barabasi and Stanley, 1995). As a result, the number of grain boundaries is decreased due to the coalescence.

Atomic Force Microscopy (AFM)

The van der Waals forces (FvdW) are interactive force acting between tip and sample was estimated by using the following relation,

$$F_{vdW} = \frac{HR}{6d^2} \dots\dots\dots(5)$$

H is the Hamaker constant and is in the order of 10^{-19} J, and R is the tip radius, and d is the spacing between tip and sample. The van der Waals forces are significantly affected by the medium in the gap between tip and sample.

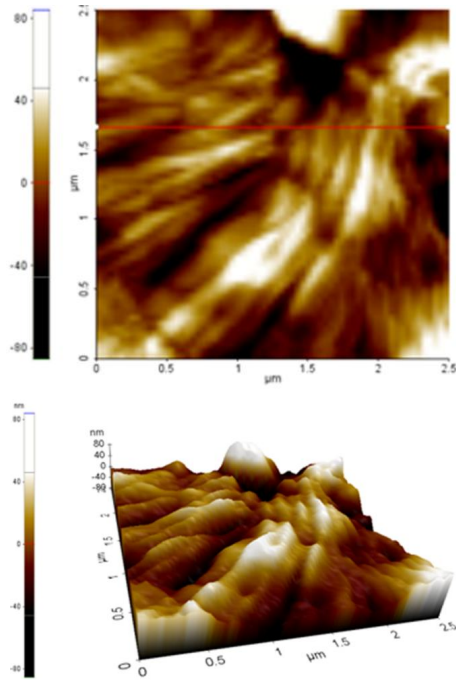


Figure 4.1. AFM 2-D & 3-D images of lead films annealed at 543K for 30 min

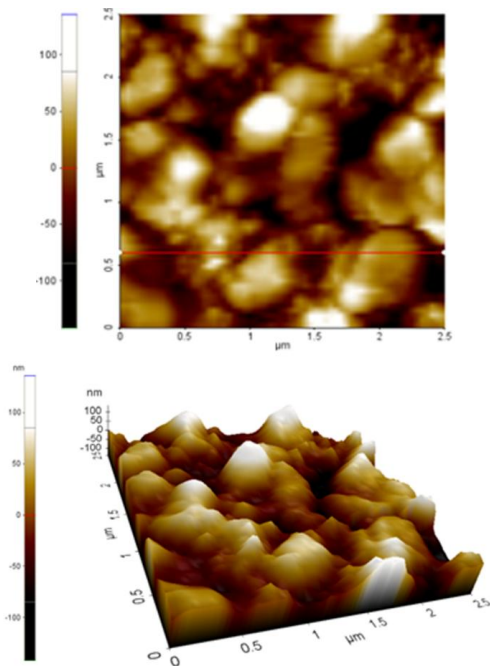


Figure 4.2. AFM 2-D & 3-D images of lead films annealed at 543K for 60 min

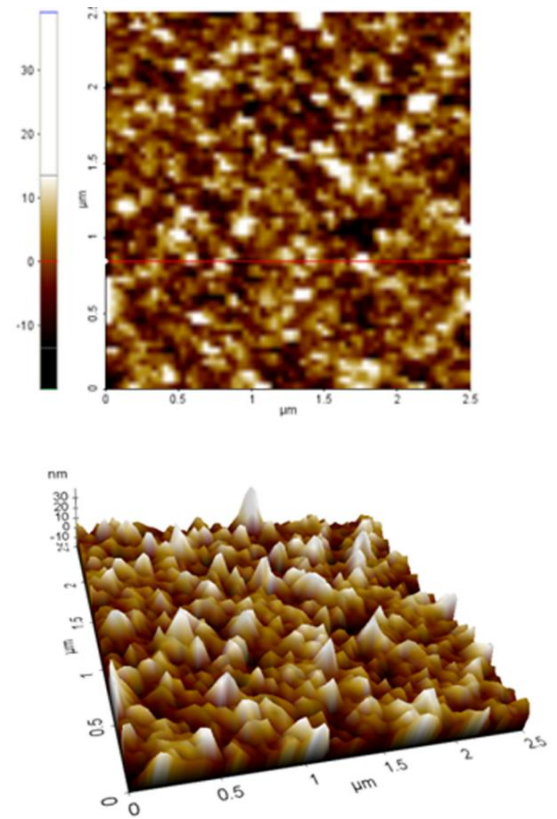


Figure 4.3. AFM 2-D & 3-D images of lead films annealed at 543K for 120 min

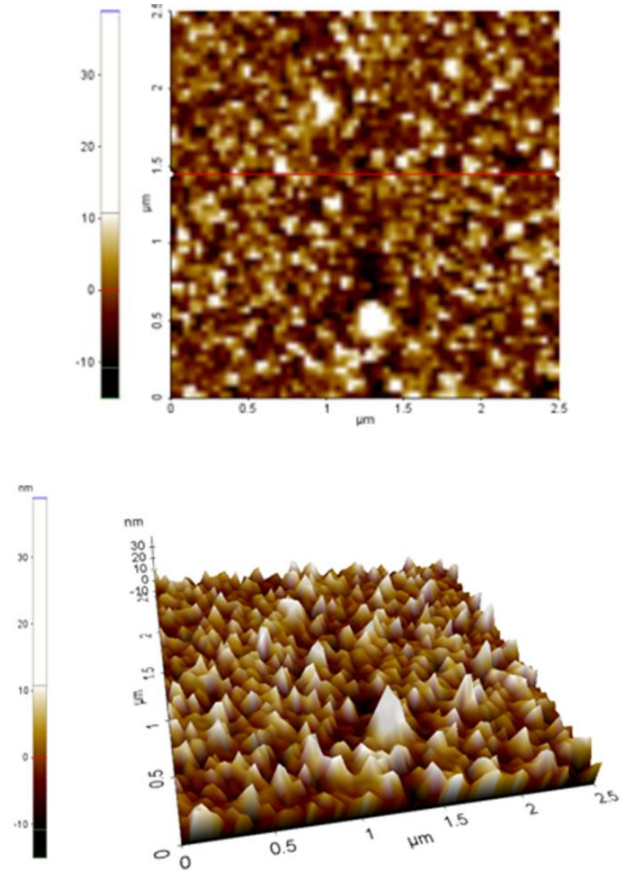


Figure 4.4. AFM 2-D& 3-D images of lead films annealed at 543K for 180 min

The results of an AFM surface roughness analysis of the lead oxide thin films over a scanning area, 2.5 μm x 2.5 μm are shown in Fig 4. Fig 4.1 – 4.4 shows the 2-dimensional surfaces of lead oxide thin films, and three-dimensional topography of lead oxide films, where roughness are seen with well-defined grains. The roughness gives an impressive insight of the excellent capability of the method to measure surface topography. To study the changes in roughness of the film, it is very important that the area analysed should remain well adhered to the substrate. It is also important that the area chosen to be the same for all measurements. In order to quantitatively describe the surface morphology, Scaling theory (Family and Vicsek, 1991; Barabasi and Stanley, 1995) was used to analyse the quantitative information on the surface morphology of AFM data. The amplitude parameters like R_a , R_q and R_{max} are used to evaluate the surface morphological properties of the thin films while functional parameters (R_{skw} , R_{kur}) are used to evaluate the surface topography. In this work both amplitude and functional parameters are calculated as a function of annealing time and are given in Table 2.

Table 2. Analysis of the AFM profile of lead oxide thin films

Annealing time (min)	R_{p-v} (nm)	(R_q) (nm)	(R_a) (nm)	R_{skw}	R_{kur}	R_z (nm)
30	50.87	11.34	9.23	0.38	2.73	31.30
60	113.15	31.36	27.00	0.24	1.85	77.05
120	41.58	8.08	6.49	-0.78	3.64	25.76
180	30.83	5.75	4.17	-1.40	5.60	21.65

The Skewness and kurtosis are used to measure symmetry and peaks of the surface (Family *et al.*, Barabasi *et al.*, 1991). The Skewness (R_{skw}) of the lead oxide films is close to 0 and kurtosis (R_{kur}) lies in the range of 1.8 to 5.6. It indicates that, the surfaces exhibited an asymmetrical distribution of grain and irregular shaped peaks and valleys. The AFM results are in good agreement with the SEM and XRD results.

Optical properties

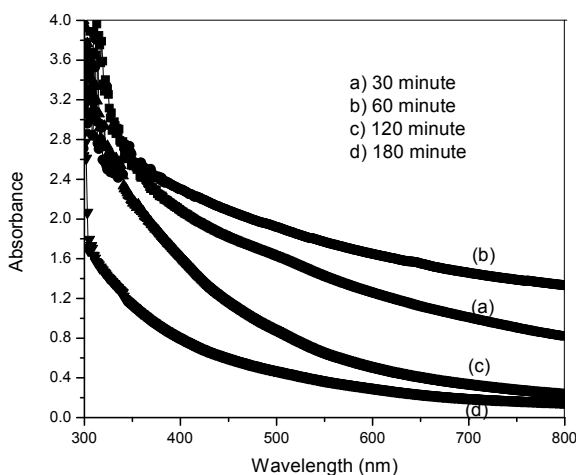


Figure 5.1. Plot of optical absorption versus wavelength for lead oxide films

Fig. 5.1 shows the variation of optical absorption A with wavelength λ. All the films show lower absorption in the

visible wavelength region. The nature of the transition (direct or indirect) was determined by the relation.

$$\alpha hv = A(hv - E_g)^n \dots\dots\dots(6)$$

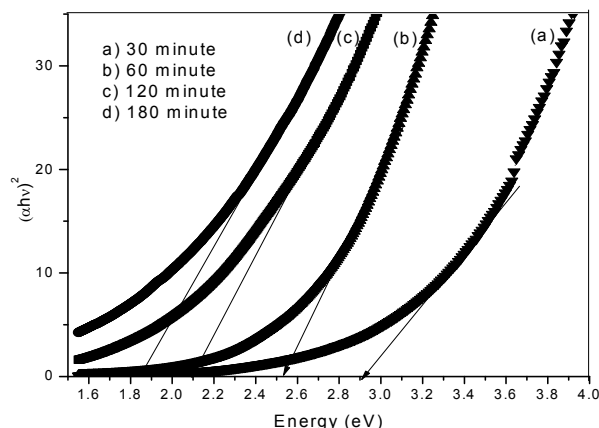


Figure 5.2. Plot of $(\alpha hv)^2$ versus Energy for lead oxide films

From the transmittance data and the Tauc’s (Tauc, 1974) relation, allowed direct transitions $((\alpha hv)^2)$ versus incident photon energy (E), plots were obtained. The allowed direct transitions ($n = 1/2$) curve is shown in Fig. 5.2. The band gap energy E_g was determined by extrapolating the tangent line portion to the x-axis for zero absorption coefficients α, and the values of E_g are in a good agreement with other reports (Droessler *et al.*, 2012). The calculated optical energy band gap was in between 2.92eV and 1.84eV.

The extinction coefficient (or) total attenuation coefficient kw was employed from the relation (Mahmoud *et al.*, 2009).

$$k = \alpha \lambda / 4\pi \dots\dots\dots(7)$$

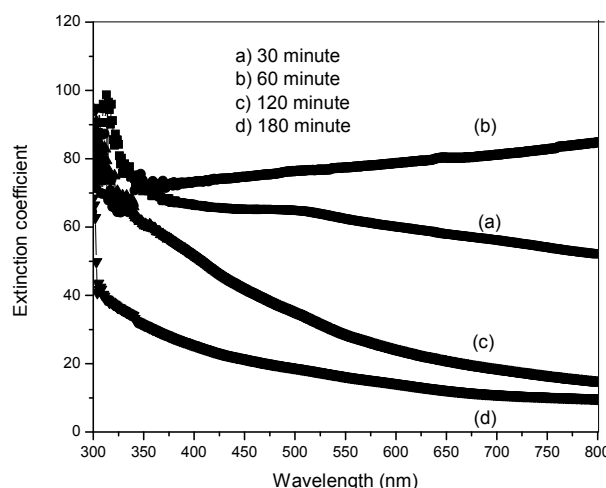


Figure 5.3. Extinction coefficient versus Wavelength for lead oxide films

The results of wavelength dependent extinction coefficient k of lead oxide thin films oxidized at 543K for different annealing time were illustrated in Fig 5.3. It is clearly exhibited that the

extinction coefficient k decreases with increasing annealing time. The observed k values decrease in the long wavelength range (Ubale and Kulkarni, 2006). The decrease in k with increasing substrate temperature might be due to the improvement in the crystallinity which leads to minimum imperfection.

Fourier Transform Infrared Spectroscopy (FTIR)

The FTIR spectroscopy is an analytical tool for the characterization of functional group of thin films. The identified possible modes of vibration from the FTIR spectra of lead oxide thin films are shown in Fig 6.

The absorption peak nearly at 460 cm^{-1} and 658 cm^{-1} indicates the presence of symmetric bending vibration modes of Pb-O bonds and asymmetric bending vibration modes of Pb-O-Pb (Trettenhahn et al., 1993; Bosca et al., 2009). The intensity of the band at 658 cm^{-1} in the spectrum shows the increase in intensity with increasing annealing time and it strongly recommends the development of Pb-O bonds.

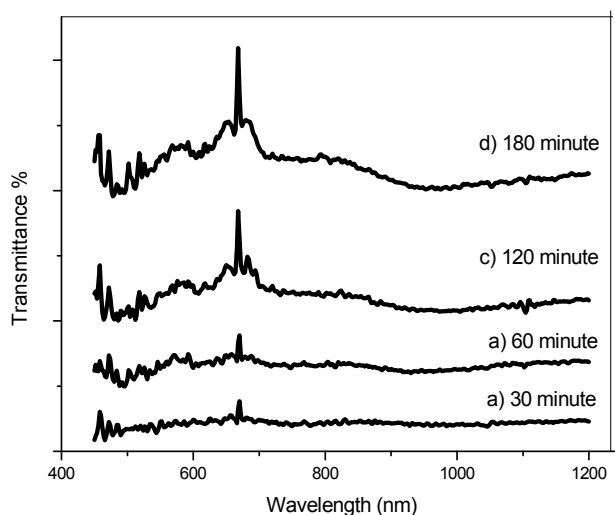


Figure 6. FT-IR Spectra of lead oxide films

Conclusion

The preparation of nano-structured lead oxide thin films has been done by a vacuum thermal evaporation method. The x-ray diffraction study shows that, when the oxidation time increases, the observed Pb_2O_3 peaks shift to PbO peak with sharp intensity and reach mono phase PbO at 180 min. The surface characterization of these films studied using scanning electron microscopy and atomic force microscopy, which reveals that uniform coating renders spherical structures with lower surface roughness. The optical absorption study revealed that, the lead oxide thin films, have allowed direct transitions. All the observed characteristic absorption peaks of the FTIR spectrum confirmed the presence of PbO. The optical energy band gap varied from 2.92eV to 1.84eV with oxidized time and this optical energy band gap is suitable for photovoltaic materials.

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